

**Table 3.** Coordination number CN, values of the local error parameter  $E_p$ , the local average dihedral angles  $\theta_{m,p}$  and the minimum and maximum dihedral angle of the tetrahedra in the five distinct coordination polyhedra

	CN	$E_p$	$\theta_{m,p}$ (°)	$\theta_{\min}$ (°)	$\theta_{\max}$ (°)
Co1	12	0.0623	70.571	59.288	79.972
M1	12	0.0596	70.567	60.319	77.132
Nb1	15	0.0649	70.515	54.748	79.972
Nb2	16	0.0634	70.536	59.288	77.132
Nb3	14	0.0663	70.486	54.748	79.972

**Table 4.** Values for the error parameter E and the average dihedral angle  $\theta_m$  of selected Frank-Kasper phases

Phase	Type	$E \times 10^{-2}$	$\theta_m$ (°)
Cr <sub>3</sub> Si <sup>a</sup>	A15	0.0631	70.435
Al <sub>3</sub> Zr <sub>4</sub> <sup>b</sup>	Z	0.0612	70.500
Fe <sub>7</sub> W <sub>6</sub> <sup>c</sup>	$\mu$	0.0658	70.540
Co <sub>6.30</sub> Nb <sub>6.70</sub>	$\mu$	0.0633	70.540
Cu <sub>2</sub> Mg <sup>d</sup>	C15	0.0621	70.588

Calculations are based on crystal data as reported by (a) Boren (1933); (b) Wilson, Thomas & Spooner (1960); (c) Arnfelt & Westgren (1935), and (d) Laves & Witte (1935).

The quality of the crystal was checked by precession photographs (Mo  $K\alpha$ ). The structure of the observe-reverse twinned crystal was refined with the program *SHELXL93* (Sheldrick, 1993). The geometrical analyses of the tetrahedral close (german: *dicht*) packed structures were performed with the program *TDP* (Wagner, 1994). The error parameters  $E$  were calculated using the algorithm of Brandon, Chieh, Pearson & Riley (1975).

Data collection: CAD-4 software. Cell refinement: CAD-4 software. Data reduction: *SDP* (Frenz, 1978). Program(s) used to refine structure: *SHELXL93* (Sheldrick, 1993). Molecular graphics: *SCHAKAL* (Keller, 1992).

Financial support of this work by the Deutsche Forschungsgemeinschaft and the Fonds der Chemischen Industrie is acknowledged.

Lists of structure factors and anisotropic displacement parameters have been deposited with the IUCr (Reference: DU1106). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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## A Novel Semiconducting Perovskite-Related Phase: Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17</sub>

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(Received 30 August 1994; accepted 19 January 1995)

## Abstract

The structure of strontium niobium oxide, Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17</sub>, prepared by the floating-zone melting technique, has been determined by single-crystal X-ray analysis. The structure contains distorted perovskite-type slabs parallel to (100) formed from five NbO<sub>6</sub> octahedra and Sr atoms. The Nb—O distances range from 1.89 (2) to 2.10 (2) Å for Nb1, from 1.809 (5) to 2.282 (6) Å for Nb2, and from 1.851 (6) to 2.167 (6) Å for Nb3. The NbO<sub>6</sub> conformations are (4+1+1) distorted. Sr2 and Sr3 are coordinated by 12 O atoms with Sr—O distances in the range 2.64 (1) to 3.04 (1) Å for Sr2 and 2.50 (1) to 3.06 (1) Å for Sr3. Sr1, placed near the boundary of the slab, is irregularly coordinated by five O atoms in the same slab and two O atoms in the neighbouring slab [Sr—O 2.46 (1) to 2.915 (6) Å] and by a further three O atoms at long coordination distances [3.345 (8) to 3.544 (9) Å]. The title compound is isotypic with La<sub>5</sub>Ti<sub>5</sub>O<sub>17</sub> and both structures belong to the homologous

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series  $A_{n+1}B_{n+1}O_{3n+5}$  with  $n = 4$ , being diamagnetic ( $B = \text{Nb}$ ) or paramagnetic ( $B = \text{Ti}$ ) electrically conducting compounds.

### Comment

The systems SrNbO<sub>x</sub>, CaNbO<sub>x</sub> ( $3.40 < x < 3.50$ ) and LaTiO<sub>x</sub> ( $3.00 < x < 3.50$ ) have been studied recently because of their interesting physical and structural properties, which are tunable by changing only the oxygen content,  $x$  (Lichtenberg, Widmer, Bednorz, Williams & Reller, 1991a; Lichtenberg, Williams, Reller, Widmer & Bednorz, 1991b; Williams, Schmalke, Reller, Lichtenberg, Widmer & Bednorz, 1991; Schmalke, Williams, Reller, Linden & Bednorz, 1993; Williams, Lichtenberg, Widmer, Bednorz & Reller, 1993). These perovskite-related compounds are members of the homologous series  $A_{n+1}B_{n+1}O_{3n+5}$  ( $A = \text{Sr}, \text{Ca}, \text{La}$  and  $B = \text{Nb}, \text{Ti}$ ), where  $n$  is merely determined by the oxygen content  $x$  (Fig. 1). The structure type for  $x = 3.40$  is known from the electrically non-conducting La–Ca–Ti–O and Na–Ca–Nb–O systems in which Ti and Nb are in their highest oxidation state ( $\text{Ti}^{4+}, 3d^0$  and  $\text{Nb}^{5+}, 4d^0$ ) (Portier, Fayard, Carpy & Galy, 1974; Nanot, Queyroux & Gilles, 1973).

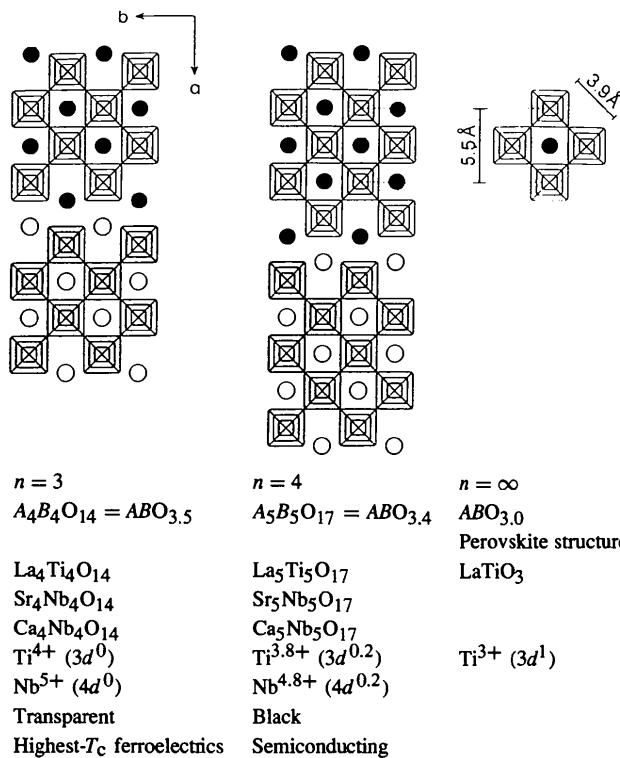


Fig. 1. Projection along the  $c$  axis of the idealized structure of the  $n = 3, 4$  and  $\infty$  members of the homologous series  $A_{n+1}B_{n+1}O_{3n+5}$ . The cations La, Sr and Ca are represented by circles. Light and heavy ruling of the  $BO_6$  octahedra and filled and open circles indicate a height difference along the projection direction.

This work describes the structure of Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17</sub> ( $x = 3.40, n = 4$ ), which is a diamagnetic electrically conducting compound ( $\text{Nb}^{4.8+}, 4d^{0.2}$ ). It was prepared by the floating-zone melting technique under an Ar atmosphere. The true stoichiometry was determined by thermogravimetric analysis to be Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17.1</sub> ( $x = 3.42$ ). It was found that for  $3.40 < x < 3.42$  the structure type ( $n = 4$ ) remains unchanged (Lichtenberg *et al.*, 1991b).

The conditions for systematic absences, taken from Weissenberg and precession photographs, led to the possible space groups  $Pnnm$  (No. 58) and  $Pnn2$  (No. 34). Statistics based on  $E$  values showed hypercentricity of the data (e.g.  $E^6 = 37.228$ , compared to theoretical values for non-centrosymmetric, centrosymmetric and hypercentric data of 6.00, 15.00 and 37.50, respectively) and the  $E^2$  values of the eight parity groups are far from unity (by definition  $\langle E^2 \rangle = 1.0$ ). Reflection groups with  $h + k + l = 2n$  have higher values than those with  $h + k + l = 2n + 1$ , influenced by superposition effects of heavy atoms along the  $a$  direction. Average values are  $eee = 2.443$ ,  $eoo = 2.909$ ,  $ooo = 0.675$ ,  $oeo = 0.497$ ;  $oee, eoe, eeo$  and  $ooo$  ( $e = \text{even}, o = \text{odd } h, k, l$ ) showed values between 0.303 and 0.364;  $\langle E^2 \rangle$  for all reflections was calculated to be 0.971.

The structure was solved in both space groups using Patterson interpretation routines of SHELXS86 (Sheldrick, 1985) and subsequent difference Fourier calculations and refinement were carried out using SHELX76 (Sheldrick, 1976). A comparison of final  $R$  ( $wR$ ) values [0.0461 (0.0601) for  $Pnnm$  and 0.0387 (0.0449) for  $Pnn2$ ] favoured the non-centrosymmetric space group  $Pnn2$ . The absolute-structure parameter  $\chi$  (Flack, 1983; Bernardinelli & Flack, 1987) refined to  $-0.025$  (32) using the  $\gamma$ -test version of SHELXL92 (Sheldrick, 1992), providing evidence that no inversion twinning was present in the crystal studied.

The atomic numbering scheme and perovskite-block separation are shown in Fig. 2. The  $\text{NbO}_6$  octahedra have average apical distances of 1.851 and 2.184 Å. [The apical O atoms are defined by the shortest and longest Nb–O distances in *trans* positions: for Nb1

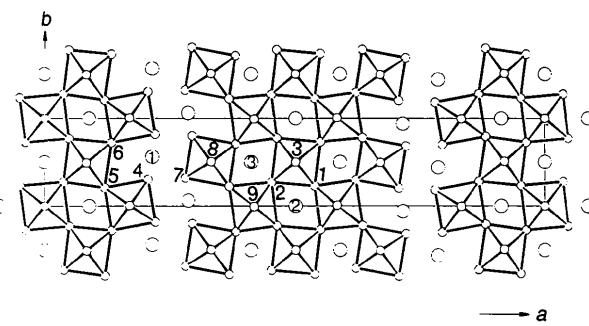


Fig. 2. A [001] projection of the crystal structure of Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17</sub>. Circled numbers 1, 2, and 3 refer to the refined Sr-atom positions, other numbers correspond to the refined O-atom positions shown in Table 1.

the apical O atoms are O<sub>3</sub> and O<sub>3</sub><sup>ii</sup>, for Nb<sub>2</sub> they are O<sub>4</sub><sup>v</sup> and O<sub>6</sub><sup>iii</sup>, and for Nb<sub>3</sub> they are O<sub>1</sub><sup>iv</sup> and O<sub>6</sub><sup>iii</sup> (see also Table 2).] The mean value of the 12 equatorial Nb—O distances for the three independent octahedra is 2.005 Å. Like the NbO<sub>6</sub> octahedra in Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> (Ishizawa, Marumo, Kawamura & Kimura, 1975) and the TiO<sub>6</sub> octahedra in the structure of La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> (Gasperin, 1975; Schmalle *et al.*, 1993), the NbO<sub>6</sub> conformations of the title compound are also (4+1+1) distorted. The NbO<sub>6</sub> octahedral blocks are shifted by a c/2 crystallographic shear, a term first used by Wadsley (1964) and reviewed by Anderson & Hyde (1989) (see the *ac* projection of the structure deposited with the supplementary material).

The Sr—O coordination distances vary considerably; the coordination spheres are shown in Fig. 3. The interplanar Sr<sub>1</sub> cation is coordinated by seven O atoms at distances between 2.46 (1) and 2.915 (6) Å, but three further O atoms, O<sub>6</sub>, O<sub>6</sub><sup>ii</sup>, and O<sub>8</sub><sup>iv</sup>, at distances between 3.35 (1) and 3.54 (1) Å (Table 2), are also indicated in Fig. 3. The intraplanar Sr<sub>2</sub> and Sr<sub>3</sub> cations have cubo-octahedral coordination spheres in the form of cubes of eight O atoms capped by the four O atoms O<sub>3</sub>, O<sub>3</sub><sup>ix</sup>, O<sub>9</sub> and O<sub>9</sub><sup>v</sup> for Sr<sub>2</sub> and the four O atoms O<sub>3</sub>, O<sub>8</sub>, O<sub>9</sub> and O<sub>9</sub><sup>x</sup> for Sr<sub>3</sub>, with Sr—O distances up to 3.04 (1) Å for Sr<sub>2</sub> and 3.06 (1) Å for Sr<sub>3</sub>. The coordination number is obviously not given correctly by the sum of the covalent radii of Sr and O (2.81 Å). If this distance were taken as a borderline criterion, Sr<sub>1</sub> and Sr<sub>2</sub> would be only coordinated by six and Sr<sub>3</sub> by seven O atoms, which is obviously not the case for this structure.

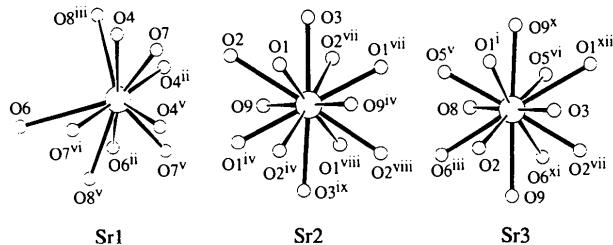


Fig. 3. Ball-and-stick perspective SCHAKAL (Keller, 1986) plot of the three Sr—O coordination spheres of Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17</sub>. The O-atom numbering corresponds to that in Table 2.

## Experimental

Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17</sub> was prepared by the floating-zone melting technique under an Ar atmosphere.

### Crystal data

Sr<sub>5</sub>Nb<sub>5</sub>O<sub>17</sub>  
*M*<sub>r</sub> = 1174.62  
 Orthorhombic  
*Pnn2*  
*a* = 32.456 (5) Å  
*b* = 5.674 (2) Å  
*c* = 3.995 (2) Å

Mo  $K\alpha$  radiation  
 $\lambda$  = 0.71073 Å  
 Cell parameters from 25 reflections  
 $\theta$  = 5.6–12.0°  
 $\mu$  = 21.11 mm<sup>-1</sup>  
 $T$  = 296 K

*V* = 735.7 (6) Å<sup>3</sup>  
 $Z$  = 2  
 $D_x$  = 5.302 Mg m<sup>-3</sup>

Ledge-shaped  
 0.387 × 0.135 × 0.035 mm  
 Black

### Data collection

Enraf–Nonius CAD-4 diffractometer

$\omega$ -2θ scans

Absorption correction:  
 analytical (based on six crystal faces)  
 $T_{\min}$  = 0.0717,  $T_{\max}$  = 0.4744

2674 measured reflections  
 2118 independent reflections

1796 observed reflections [ $|F| > 3\sigma(F)$ ]

$R_{\text{int}}$  = 0.0581

$\theta_{\text{max}}$  = 30.0°

$h$  = -45 → 45

$k$  = 0 → 8

$l$  = -5 → 0

3 standard reflections  
 frequency: 180 min  
 intensity decay: <1%

### Refinement

Refinement on *F*

$R$  = 0.0387

$wR$  = 0.0449

$S$  = 3.396

1785 reflections

80 parameters

$w = 8.826/[\sigma^2(F_o) + 0.000010F_o^2]$

( $\Delta/\sigma$ )<sub>max</sub> = 0.116  
 $\Delta\rho_{\text{max}}$  = 2.19 e Å<sup>-3</sup>

$\Delta\rho_{\text{min}}$  = -3.09 e Å<sup>-3</sup>

Atomic scattering factors  
 from International Tables  
 for X-ray Crystallography  
 (1974, Vol. IV)

Table 1. Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters (Å<sup>2</sup>)

$$U_{\text{iso}}$$
 for O atoms,  $U_{\text{eq}} = (1/3)\sum_i \sum_j U_{ij} a_i^* a_j^* \mathbf{a}_i \cdot \mathbf{a}_j$  for Nb and Sr.

	<i>x</i>	<i>y</i>	<i>z</i>	$U_{\text{eq}}/U_{\text{iso}}$
Nb1	1/2	1/2	0.5	0.0035 (2)
Nb2	0.32270 (2)	0.5339 (1)	0.5051 (12)	0.0044 (1)
Nb3	0.40928 (2)	0.0053 (1)	0.5120 (12)	0.0031 (1)
Sr1	0.21443 (2)	0.5628 (1)	0.5130 (12)	0.0136 (2)
Sr2	1/2	0	-0.0010 (14)	0.0068 (2)
Sr3	0.41131 (3)	0.4996 (1)	0.0027 (11)	0.0072 (2)
O1	0.5376 (2)	0.2258 (8)	0.4890 (37)	0.009 (1)
O2	0.4518 (2)	0.2693 (8)	0.5267 (33)	0.006 (1)
O3	1/2	1/2	0.0264 (41)	0.015 (2)
O4	0.2089 (2)	0.2966 (9)	0.0153 (39)	0.009 (1)
O5	0.1207 (2)	0.2070 (8)	0.0171 (37)	0.006 (1)
O6	0.1328 (2)	0.7236 (9)	-0.0300 (28)	0.010 (1)
O7	0.2794 (2)	0.3177 (9)	0.5073 (40)	0.007 (1)
O8	0.3311 (2)	0.5054 (10)	0.0196 (38)	0.017 (2)
O9	0.4169 (2)	-0.0108 (10)	0.0138 (38)	0.015 (1)

Table 2. Selected geometric parameters (Å, °)

Nb1—O <sub>3</sub>	1.892 (16)	Sr2—O <sub>1</sub> <sup>viii</sup>	2.699 (13)
Nb1—O <sub>1</sub>	1.978 (5)	Sr2—O <sub>9</sub>	2.699 (7)
Nb1—O <sub>1</sub> <sup>i</sup>	1.978 (5)	Sr2—O <sub>9</sub> <sup>iv</sup>	2.699 (7)
Nb1—O <sub>2</sub>	2.042 (5)	Sr1—O <sub>4</sub> <sup>v</sup>	2.821 (6)
Nb1—O <sub>2</sub> <sup>i</sup>	2.042 (5)	Sr1—O <sub>8</sub> <sup>v</sup>	2.915 (6)
Nb1—O <sub>3</sub> <sup>ii</sup>	2.103 (16)	Sr1—O <sub>6</sub> <sup>ii</sup>	3.345 (8)
Nb3—O <sub>6</sub> <sup>iii</sup>	1.851 (6)	Sr1—O <sub>8</sub> <sup>iii</sup>	3.492 (6)
Nb3—O <sub>5</sub> <sup>iii</sup>	1.952 (5)	Sr1—O <sub>6</sub>	3.544 (9)
Nb3—O <sub>9</sub>	2.007 (16)	Sr2—O <sub>3</sub>	2.839 (1)
Nb3—O <sub>9</sub> <sup>ii</sup>	2.022 (16)	Sr2—O <sub>3</sub> <sup>ii</sup>	2.839 (1)
Nb3—O <sub>2</sub>	2.039 (5)	Sr2—O <sub>2</sub> <sup>vii</sup>	2.888 (10)
Nb3—O <sub>1</sub> <sup>iv</sup>	2.167 (6)	Sr2—O <sub>2</sub> <sup>viii</sup>	2.888 (10)
Nb2—O <sub>4</sub> <sup>v</sup>	1.809 (5)	Sr2—O <sub>2</sub>	3.037 (11)
Nb2—O <sub>7</sub>	1.866 (5)	Sr2—O <sub>2</sub> <sup>iv</sup>	3.037 (11)
Nb2—O <sub>8</sub>	1.966 (16)	Sr3—O <sub>5</sub> <sup>vi</sup>	2.495 (13)
Nb2—O <sub>8</sub> <sup>ii</sup>	2.080 (16)	Sr3—O <sub>5</sub> <sup>v</sup>	2.586 (13)
Nb2—O <sub>5</sub> <sup>v</sup>	2.084 (5)	Sr3—O <sub>8</sub>	2.603 (7)
Nb2—O <sub>6</sub> <sup>iii</sup>	2.282 (6)	Sr3—O <sub>2</sub> <sup>vii</sup>	2.656 (11)
Sr1—O <sub>7</sub> <sup>v</sup>	2.456 (14)	Sr3—O <sub>9</sub> <sup>x</sup>	2.784 (6)
Sr1—O <sub>7</sub> <sup>vi</sup>	2.493 (14)	Sr3—O <sub>2</sub>	2.797 (11)
Sr1—O <sub>4</sub>	2.504 (13)	Sr3—O <sub>6</sub> <sup>iii</sup>	2.826 (9)
Sr1—O <sub>4</sub> <sup>ii</sup>	2.518 (13)	Sr3—O <sub>3</sub>	2.880 (1)

Sr1—O7	2.525 (6)	Sr3—O9	2.902 (6)
Sr2—O1 <sup>iv</sup>	2.639 (12)	Sr3—O1 <sup>i</sup>	2.992 (11)
Sr2—O1 <sup>vii</sup>	2.639 (12)	Sr3—O6 <sup>xi</sup>	3.005 (10)
Sr2—O1 <sup>viii</sup>	2.699 (13)	Sr3—O1 <sup>xii</sup>	3.064 (11)
O3—Nb1—O1	88.7 (4)	O7—Nb2—O6 <sup>iii</sup>	88.3 (2)
O3—Nb1—O1 <sup>i</sup>	88.7 (4)	O8—Nb2—O8 <sup>ii</sup>	161.9 (4)
O3—Nb1—O2	93.0 (4)	O8—Nb2—O5 <sup>v</sup>	86.5 (4)
O3—Nb1—O2 <sup>i</sup>	93.0 (4)	O8—Nb2—O6 <sup>iii</sup>	77.7 (4)
O3—Nb1—O3 <sup>ii</sup>	180.0	O8 <sup>ii</sup> —Nb2—O5 <sup>v</sup>	84.1 (5)
O1—Nb1—O1 <sup>i</sup>	177.8 (5)	O8 <sup>ii</sup> —Nb2—O6 <sup>iii</sup>	85.3 (3)
O1—Nb1—O2	88.2 (2)	O5 <sup>v</sup> —Nb2—O6 <sup>iii</sup>	78.8 (2)
O1—Nb1—O2 <sup>i</sup>	91.9 (2)	O6 <sup>iii</sup> —Nb3—O5 <sup>iii</sup>	102.3 (2)
O1—Nb1—O3 <sup>ii</sup>	91.3 (4)	O6 <sup>iii</sup> —Nb3—O9	91.8 (4)
O1 <sup>i</sup> —Nb1—O2	91.9 (2)	O6 <sup>iii</sup> —Nb3—O9 <sup>ii</sup>	102.1 (4)
O1 <sup>i</sup> —Nb1—O2 <sup>i</sup>	88.2 (2)	O6 <sup>iii</sup> —Nb3—O2	90.6 (2)
O1 <sup>i</sup> —Nb1—O3 <sup>ii</sup>	91.3 (4)	O6 <sup>iii</sup> —Nb3—O1 <sup>iv</sup>	170.9 (5)
O2—Nb1—O2 <sup>i</sup>	174.0 (7)	O5 <sup>iii</sup> —Nb3—O9	91.8 (5)
O2—Nb1—O3 <sup>ii</sup>	87.0 (4)	O5 <sup>iii</sup> —Nb3—O9 <sup>ii</sup>	90.6 (5)
O2 <sup>i</sup> —Nb1—O3 <sup>ii</sup>	87.0 (4)	O5 <sup>iii</sup> —Nb3—O2	167.0 (2)
O4 <sup>v</sup> —Nb2—O7	96.6 (2)	O5 <sup>iii</sup> —Nb3—O1 <sup>iv</sup>	82.7 (2)
O4 <sup>v</sup> —Nb2—O8	99.7 (6)	O9—Nb3—O9 <sup>ii</sup>	165.0 (4)
O4 <sup>v</sup> —Nb2—O8 <sup>ii</sup>	96.7 (6)	O9—Nb3—O2	88.8 (4)
O4 <sup>v</sup> —Nb2—O5 <sup>v</sup>	96.4 (2)	O9—Nb3—O1 <sup>iv</sup>	80.4 (4)
O4 <sup>v</sup> —Nb2—O6 <sup>iii</sup>	174.7 (3)	O9 <sup>ii</sup> —Nb3—O2	85.5 (4)
O7—Nb2—O8	93.2 (5)	O9 <sup>ii</sup> —Nb3—O1 <sup>iv</sup>	85.3 (4)
O7—Nb2—O8 <sup>ii</sup>	92.5 (5)	O2—Nb3—O1 <sup>iv</sup>	84.7 (2)
O7—Nb2—O5 <sup>v</sup>	166.9 (2)		

Symmetry codes: (i)  $1-x, 1-y, z$ ; (ii)  $x, y, 1+z$ ; (iii)  $\frac{1}{2}-x, y-\frac{1}{2}, \frac{1}{2}+z$ ; (iv)  $1-x, -y, z$ ; (v)  $\frac{1}{2}-x, \frac{1}{2}+y, \frac{1}{2}+z$ ; (vi)  $\frac{1}{2}-x, \frac{1}{2}+y, z-\frac{1}{2}$ ; (vii)  $x, y, z-1$ ; (viii)  $1-x, -y, z-1$ ; (ix)  $x, y-1, z$ ; (x)  $x, 1+y, z$ ; (xi)  $\frac{1}{2}-x, y-\frac{1}{2}, z-\frac{1}{2}$ ; (xii)  $1-x, 1-y, z-1$ .

Data collection, cell refinement and crystal-face indexing: CAD-4 Software (Enraf–Nonius, 1989). Data reduction: MolEN (Fair, 1990). Structure solution: SHELXS86 (Sheldrick, 1985). Structure refinement and absorption correction: SHELX76 (Sheldrick, 1976). 12 reflections affected by extinction were excluded from the final cycles of least-squares refinement. Anisotropic displacement factors were refined only for Sr and Nb atoms. Molecular graphics: SCHAKAL (Keller, 1986). Geometry calculations: ORFFE3 (Busing *et al.*, 1971).

Gratitude is expressed to Professor H. R. Oswald, the University of Zürich, for standing support for this work, and the Swiss National Foundation of Science for financial support under project NF 2.772-0.87. The authors also thank the coeditor, Dr I. David Brown, for helpful suggestions and for improving the style of the paper.

Lists of structure factors, anisotropic displacement parameters and complete geometry, and a plot of the projection of the structure on the *ac* plane, have been deposited with the IUCr (Reference: BR1096). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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*Acta Cryst.* (1995). **C51**, 1246–1249

## A New Type of Mixed Anionic Framework in the Structure of K<sub>3</sub>[Al<sub>4</sub>F<sub>9</sub>(PO<sub>4</sub>)<sub>2</sub>]

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(Received 8 November 1994; accepted 11 January 1995)

## Abstract

X-ray analysis of a hydrothermally synthesized potassium aluminium fluoride orthophosphate, K<sub>3</sub>[Al<sub>4</sub>F<sub>9</sub>(PO<sub>4</sub>)<sub>2</sub>], showed that this compound has a new type of